



# Insights into enhancement of NH<sub>3</sub>-SCR activity and N<sub>2</sub> selectivity of LDHs-derived NiMnAlO<sub>x</sub> catalysts: Combination of experiments and DFT calculations

Qinghua Yan<sup>a,1</sup>, Jiewen Xiao<sup>b,1</sup>, Rongrong Gui<sup>b</sup>, Zhenyu Chen<sup>a</sup>, Yuxin Wang<sup>a</sup>, Yuran Li<sup>c</sup>, Tingyu Zhu<sup>c</sup>, Qiang Wang<sup>b,\*</sup>, Yanjun Xin<sup>a,\*\*</sup>

<sup>a</sup> Qingdao Engineering Research Center for Rural Environment, College of Resources and Environment, Qingdao Agricultural University, Qingdao 266109, PR China

<sup>b</sup> College of Environmental Science and Engineering, Beijing Forestry University, 35 Qinghua East Road, Haidian District, Beijing 100083, PR China

<sup>c</sup> Research Center for Process Pollution Control, National Engineering Laboratory for Hydrometallurgical Cleaner Production Technology, Institute of Process Engineering, Chinese Academy of Sciences, Beijing 100190, China

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## ABSTRACT

The LDHs-derived NiMnAlO<sub>x</sub> catalysts was proposed for NH<sub>3</sub>-SCR denitration technology, which improved the overall catalytic performance. At 100–250 °C, the preferred Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> had NO<sub>x</sub> conversion of exceeded 95%, and N<sub>2</sub> selectivity of more than 89%, which was higher than the control catalysts. Additionally, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> also performed better resistance to SO<sub>2</sub> with less sulfate species deposited on the surface. Transient reaction was conducted to illustrate that Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> reduced the N<sub>2</sub>O formation by suppressing the occurrence of non-selective catalytic pathway. Physicochemical characterization analyses confirmed that Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst had more active species, abundant surface oxygen, moderate acidic sites and redox properties, which promoted the adsorption and activation of NO<sub>x</sub> and NH<sub>3</sub>, and further accelerated the SCR reaction. Finally, DFT calculations provided insights into the mechanism and influence of the degradation of the Jahn-Teller effect and the reversal of the magnetic moment at the Mn site on the surface adsorption process.

## 1. Introduction

Nitrogen oxides (NO<sub>x</sub>) are one of the major pollutants emitted from industrial production, and their harmful effects on the ecosystem and human health cannot be ignored [1,2]. Compared with the existing commercial catalytic systems (V<sub>2</sub>O<sub>5</sub>-WO<sub>3</sub>(MoO<sub>3</sub>)/TiO<sub>2</sub>) that suffer from decreased N<sub>2</sub> selectivity at high temperatures (573–673 K), high toxicity of V<sub>2</sub>O<sub>5</sub>, high conversion rate of SO<sub>2</sub> to SO<sub>3</sub>, and the need to reheat stack gases [3], the low-temperature selective catalytic reduction (LT-SCR) of NO<sub>x</sub> technology has been favored by both scholars and industries. Therefore, from the perspective of energy crisis and global climate deterioration, excellent LT-SCR denitration catalysts are more suitable for practical applications.

In the past decades, large amount of catalysts have been developed for LT-SCR, such as Ce-[4–7], Cu-[8], Fe-[6,9,10], Mn-[11,12] based catalysts. Especially, Mn-based catalysts, possessing a various unstable

valence states and remarkable redox properties, have been the hotspot for LT-SCR of NO<sub>x</sub>. Although LT-SCR has been studied quite intensively, most of the studies have focused mainly on the conversion of NO, with little attention paid to the conversion of N<sub>2</sub>O. The amount of N<sub>2</sub>O produced by V<sub>2</sub>O<sub>5</sub> based catalysts is usually small, but considerable formation of N<sub>2</sub>O has been detected in SCR processes using Mn-based catalysts at low temperature. For SCR technology, N<sub>2</sub>O from NO conversion is another form of pollution and due to its high stability, it is more effective than CO<sub>2</sub> and CH<sub>4</sub> in causing stratospheric ozone depletion. In addition, it has been shown that the high performance of catalysts even at high temperatures above 300 °C may be attributed to the decomposition of N<sub>2</sub>O. Therefore, exploring the pathway and mechanism of N<sub>2</sub>O generation is crucial for the development of Mn-based catalysts with overall excellent performance.

Generally, N<sub>2</sub>O can generate from the direct oxidation of NH<sub>3</sub> [13,14], the non-selective catalytic reaction [15,16], and the NO

\* Correspondence to: Beijing Forestry University, PR China.

\*\* Correspondence to: Qingdao Agricultural University, PR China.

E-mail addresses: [qiangwang@bjfu.edu.cn](mailto:qiangwang@bjfu.edu.cn) (Q. Wang), [xintom2000@126.com](mailto:xintom2000@126.com) (Y. Xin).

<sup>1</sup> These authors contributed equally.

disproportionation reaction [16]. Ren et al. [17] found that the  $\text{N}_2\text{O}$  accumulation on the  $\text{MnFeO}_x$  catalyst at low temperature ( $<150^\circ\text{C}$ ) mainly came from the non-selective catalytic reaction pathway, and then gradually shifted to  $\text{NH}_3$  over-oxidation with the increase of temperature. Moreover,  $\text{NH}_2\text{NO}$  and  $\text{NH}_4\text{NO}_2$  intermediates played a vital part in  $\text{NH}_3$ -SCR reaction and prevented the generation of  $\text{N}_2\text{O}$  [18]. Metal element doping modification is undoubtedly an efficient strategy for improving the overall performance of Mn-based SCR catalysis [19, 20]. Chen et al. [21] reported that the synergistic effect of Ni and Mn formed by electron transfer can improve the redox ability of the Mn-based catalyst, thereby promoting the  $\text{DeNO}_x$  activity and  $\text{N}_2$  selectivity. The contribution value of the above reaction path for the formation of  $\text{N}_2\text{O}$ , and the essential change process of reaction molecules for Mn-based catalysts were not yet very clear. Wan et al. [22] reported that the preferred  $\text{Ni}(0.4)\text{-MnO}_x$  catalyst reached nearly 100%  $\text{NO}_x$  conversion in the  $\text{NH}_3$ -SCR of  $\text{NO}_x$  at  $150\text{--}240^\circ\text{C}$ , and possessed certain resistance to  $\text{SO}_2$ . However, its  $\text{N}_2$  selectivity was still unsatisfactory, only 70% at low temperatures. Deng et al. [23] confirmed that the introduction of Fe can improve the overall performance of  $\text{Fe}(0.5)\text{-MnO}_x/\text{TiO}_2$  catalyst, but its activity at low temperatures and  $\text{N}_2$  selectivity at high temperatures still need to be improved. For LT-SCR catalysts, the dispersion state of  $\text{MnO}_x$  is one of the decisive factors for catalytic activity [23]. Smirniotis et al. [24] reported that the aggregate state of  $\text{MnO}_x$  has a great influence on the performance of the catalyst. Kijlstra et al. [25] also proposed that the aggregated  $\text{MnO}_x$  phase can accelerate the production of  $\text{N}_2\text{O}$ , which affected the  $\text{N}_2$  selectivity. Therefore, the high dispersion state of the active component is the non-negligible parameter for designing an excellent LT-SCR catalyst.

However, the conventional catalyst preparation methods (e.g., impregnation and co-precipitation) lead to poor dispersion of the highly loaded amorphous Mn phase, which reduces denitrification activity [16]. Recently, SCR catalysts produced by the layered double hydroxides (LDHs) derived catalyst method have shown excellent catalytic activity, lower apparent activity energy, and better dispersion of active particles compared to most of the previously reported SCR catalysts. LDHs are a class of two-dimensional nanostructured layered anionic materials, and its laminated metal cations and interlayer anions have adjustable properties [26,27]. The LDHs material can be calcined at high temperature to obtain layered double oxides (LDOs) with large specific surface area, high dispersion of active components and controllability, which has become the potential application material in various fields, such as  $\text{CO}_2$  capture [28,29], adsorbents [30,31], catalysts [32,33], and two-dimensional membrane materials [34].

Herein, we designed the LDHs-derived  $\text{NiMnAlO}_x$  mixed oxides, which may be excellent  $\text{NH}_3$ -SCR catalysts for  $\text{NO}_x$ , especially for low-temperature activity and  $\text{N}_2$  selectivity. The high dispersibility and controllability of LDHs-derived oxides, as well as the advantage of Ni element, were fully combined to broaden the catalytic temperature window and improve  $\text{N}_2$  selectivity and resistance to  $\text{SO}_2$ . The effects of parameters such as metal composition and calcination conditions on the crystal structure, active species, redox properties, surface acidity and catalytic performance were systematically studied, and the optimal preparation process was clarified. Importantly, the transient reaction was conducted to detect the contribution of the  $\text{N}_2\text{O}$  generation pathways and explain the nature of the excellent  $\text{N}_2$  selectivity on the  $\text{NiMnAlO}_x$ -LDO catalysts. Density functional theory (DFT) was combined with in-situ DRIFTS analysis to analyze the electronic layer structure and intermolecular forces, which revealed the adsorption and activation properties of  $\text{NO}_x$  and  $\text{NH}_3$  on the Ni doped Mn-based catalysts, inhibited the overoxidation of  $\text{NH}_3$  species, and avoided the further dehydrogenation of  $\text{NH}_2$  species, thus increasing the  $\text{N}_2$  production.

## 2. Experimental

### 2.1. Catalyst preparation

$\text{Ni}_y\text{Mn}_z\text{Al}_{1-y-z}\text{-CO}_3$  LDHs ( $0 < y < 3$ ,  $0 < z < 1$ ) were prepared by the co-precipitation method. The following will take the preparation of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{-CO}_3$  LDH as an example: According to a certain ratio, 3.63 g  $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ , 1.45 mL 50%  $\text{Mn}(\text{NO}_3)_2$ , and 2.34 g  $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  were mixed and configured into a 100 mL solution, add it dropwise to the continuously stirring 100 mL  $\text{Na}_2\text{CO}_3$  (0.025 mol) solution, and keep the pH of the system at about 10 by adding 4 M NaOH solution. The temperature during the synthesis was maintained at  $60^\circ\text{C}$  and performed under exposure with air. After dripping, keep stirring for 12 h. Then, suction filtration and washing until  $\text{pH} = 7$ , and stirring the slurry in ethanol solution for 2 h. Finally, it was dried at  $60^\circ\text{C}$  and ground for use, and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{-CO}_3$  LDH was obtained.  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-y-z}\text{O}_x$  LDOs were obtained by calcination of LDHs at high temperature. In order to systematically evaluate the performance of  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-y-z}\text{O}_x$ , 2 wt% Mn/ $\gamma\text{-Al}_2\text{O}_3$  and 1 wt% Ni-1 wt% Mn/ $\gamma\text{-Al}_2\text{O}_3$  as control catalysts were obtained by the conventional impregnation method [32].

### 2.2. Catalyst characterization

X-Ray Diffraction (XRD-7000, Shimadzu), Field Emission Scanning Electron Microscope (FE-SEM, SU-8010, Hitachi), and High Resolution Transmission Electron Microscopy (HR-TEM, FEI Tecnai G2 F20, USA) were used to reveal the structural and morphological characteristics of the prepared material. The specific surface area of prepared material was analyzed by  $\text{N}_2$  sorption/desorption analysis (SSA-7000, Builder). X-ray photoelectron spectroscopy (XPS) analyses was implemented on Thermo Scientific Escalab 250Xi with monochromatic Al  $\text{K}\alpha$  radiation ( $h\nu = 1486.6\text{ eV}$ ). The total Mn content in the sample was analyzed by inductively coupled plasma mass spectrometer (ICP-MS, Agilent 7900, USA). Multifunction chemisorption analyzer was used on Builder PCA-1200 to implement  $\text{SO}_2/\text{NH}_3/\text{NO}$ -temperature programmed desorption ( $\text{SO}_2/\text{NH}_3/\text{NO}$ -TPD) and  $\text{H}_2$ -temperature programmed reduction ( $\text{H}_2$ -TPR) analyses. In-situ DRIFT analyses were achieved with the reaction cell (ZnSe window) and diffuser attachment (HARRICK) on Bruker Vertex 70 instrument.

### 2.3. Catalytic activity measurements

The denitration performance was implemented in a stainless steel reactor with a fixed bed containing about 150 mg of catalyst at a flow rate of 200 mL/min. Equal concentrations (500 ppm) of  $\text{NO}_x$  and  $\text{NH}_3$ , 5%  $\text{O}_2$ , 5%  $\text{H}_2\text{O}$  (if needed) and 100 ppm  $\text{SO}_2$  (if needed), and the carrier gas Ar together formed the simulation gas. The MultiGas 2000 FTIR Gas Analyzer was selected to monitor the concentrations on-line for 1 h. The corresponding calculation in steady state was done using Eqs. (1)–(3).

$$\text{NO}_x \text{ conversion} = \left(1 - \frac{\text{NO}_x(\text{out})}{\text{NO}_x(\text{in})}\right) \times 100\% \quad (1)$$

$$\text{N}_2 \text{ selectivity} = \left(1 - \frac{2\text{N}_2\text{O}(\text{out})}{\text{NO}_x(\text{in}) + \text{NH}_3(\text{in}) - \text{NO}_x(\text{out}) - \text{NH}_3(\text{out})}\right) \times 100\% \quad (2)$$

$$\text{NH}_3 \text{ conversion} = \left(1 - \frac{\text{NH}_3(\text{out})}{\text{NH}_3(\text{in})}\right) \times 100\% \quad (3)$$

The reaction rate constant ( $k$ ) can be expressed as a function of the  $\text{NO}_x$  conversion (%) at different reaction temperatures, using Eq. (4). The  $\text{NH}_3$ -SCR catalytic reaction of  $\text{NO}_x$  on the catalyst surface follows Arrhenius Eq. (5):

$$k = -\frac{V}{W} \times \ln(1 - x) \quad (4)$$

$$\text{Ln}k = \text{Ln}A - \frac{E_a}{RT} \quad (5)$$

Here,  $k$  ( $\text{mL} \cdot \text{g}^{-1} \cdot \text{s}^{-1}$ ) represents the reaction rate constant of  $\text{NH}_3$ -SCR reaction,  $V$  ( $\text{mL} \cdot \text{s}^{-1}$ ) represents the gas flow rate of the reaction,  $W$  (g) represents the weight,  $x$  represents the  $\text{NO}_x$  conversion (%),  $A$  represents the pre-exponential factor,  $E_a$  represents the apparent activation energy ( $\text{kJ} \cdot \text{mol}^{-1}$ ),  $R$  represents the gas constant,  $T$  represents the thermodynamic temperature (K). The reaction order of the  $\text{NH}_3$ -SCR reaction of  $\text{NO}_x$  is close to the first order for  $\text{NO}$  and the zero order for  $\text{NH}_3$ .

#### 2.4. Transient reaction and $\text{NO} + \text{O}_2$ reaction

The transient reaction was implemented at 250 °C, the amount of catalyst was 0.15 g, the total gas flow rate was 200 mL/min, and the transient reaction time of “ $\text{NH}_3 + \text{O}_2$ ” and “ $\text{NO} + \text{O}_2$ ” was 1 h. The feed gas containing 500 ppm  $\text{NH}_3$ , 5%  $\text{O}_2$ , and Ar as balance gas was firstly introduced into the reactor to simulate the  $\text{NH}_3$  oxidation reaction [13, 14]. After the reaction reached steady state, another 500 ppm  $\text{NO}_x$  was introduced into the feed gas to simulate the  $\text{NH}_3$ -SCR reaction, and keeping the concentration of other gases unchanged. The concentration of gas ( $\text{NH}_3$ ,  $\text{NO}$ ,  $\text{NO}_2$  and  $\text{N}_2\text{O}$ ) was monitored online by MultiGas 2000 analyzer.

The  $\text{NO} + \text{O}_2$  reaction was also implement at 250 °C. A mixed gas containing  $\text{NO}_x$  (500 ppm), 5%  $\text{O}_2$  and Ar was introduced into the reactor to simulate the  $\text{NO}$  disproportionation reaction [16].

#### 2.5. DFT calculations

As the result of characterized by XRD (see Section 3.5),  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  was modeled by (110) surface of  $\alpha\text{-MnO}_2$  (JCPDS No.24-0735). Because the characteristic peak of  $\text{NiO}$  phase appeared rather than  $\text{Mn}$  species, the  $\text{NiMnAlO}_x$  catalyst was constructed by  $\text{NiO}$  phase (JCPDS No. 47-1049). The (111) plane of  $\text{NiO}$  was observed by HR-TEM (see Section 3.1), so the surface of  $\text{NiMnAlO}_x$  catalyst was modeled by (111) surface of  $(2 \times 2)$   $\text{NiO}$ . For simulating the  $\text{Mn}$  active site on  $\text{NiMnAlO}_x$  catalyst, a central  $\text{Ni}$  atom on (111) surface of  $\text{NiO}$  was replaced to  $\text{Mn}$ . The dispersion was taken into account by constructing the supercell, while the interaction between  $\text{Mn}$  atoms was taken into account by using the periodicity of the model. For the  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{NiMnAlO}_x$  catalysts have the same catalytic active sites, the  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$  can be explained at same time. The Lewis acid sites of  $\text{Mn}$  on the surfaces were selected as the adsorption sites of  $\text{NO}$  and  $\text{NH}_3$ .

Computational simulations were implemented by Vienna ab-initio simulation package (VASP) with the projector augmented wave pseudo-potentials (PAW) to describe the interaction between atomic cores and valence electrons. The Perdew–Burke–Ernzerhof (PBE) functional within the generalized gradient approximation (GGA) were used to implement density functional theory (DFT) calculations [35]. Eight layered  $\text{NiO}$  (111) slab models and  $5 \times 5 \times 1$  Monkhorst Pack k-point sampling were employed to simulate the  $\text{NiO}$  surface. Four layered  $\text{MnO}_2$  (110) slab models with  $3 \times 4 \times 1$  Monkhorst Pack k-point sampling were employed to simulate the  $\text{MnO}_2$  surface. In all of the structure optimization calculations, the bottom two layers were fixed, while the other atoms were fully relaxed. The reasonable vacuum layers were set around 15 Å in the z-directions for avoiding interaction between planes. A cutoff energy of 400 eV was provided was chosen for the well converged energy values. Geometry optimizations were pursued until the force on each atom falls below the convergence criterion of 0.02 eV/Å and energies were converged within  $10^{-6}$  eV. The spin polarized method was applied and the magnetic moments were calculated for all magnetic structures. Ferromagnetic and antiferromagnetic  $\text{NiO}$  bulk cells were calculated, and the lower energy anti-ferromagnetism was chosen for the magnetic calculations of  $\text{NiO}$  (111). Ferromagnetism was used in the magnetic calculations for  $\text{MnO}_2$ .

The Hubbard corrections (DFT+U) of  $\text{Mn}$  and  $\text{Ni}$  were  $U_{\text{eff}} = 2.8$  eV ( $U = 4.0$  eV,  $J = 1.2$  eV) and  $U_{\text{eff}} = 5.0$  eV ( $U = 5.0$  eV,  $J = 0$  eV).

The adsorption ability was evaluated using the adsorption energy term ( $E_{\text{ads}}$ ), which was calculated as follows:

$$E_{\text{ads}} = E_{\text{NO/NH}_3+\text{surface}} - (E_{\text{surface}} + E_{\text{NO/NH}_3}) \quad (6)$$

$E_{\text{surface}}$  represents the material surface energy,  $E_{\text{NO/NH}_3+\text{surface}}$  represents the energy of the surface of the material after absorption of  $\text{NO}$  or  $\text{NH}_3$  molecules, and  $E_{\text{NO/NH}_3}$  represents the energy of isolated  $\text{NO}$  or  $\text{NH}_3$  molecule.

The d band center ( $\epsilon_d$ ) [36] of material was evaluated as follows:

$$\epsilon_d = \frac{\int_{-\infty}^{\infty} n_d(\epsilon) \epsilon d\epsilon}{\int_{-\infty}^{\infty} n_d(\epsilon) d\epsilon} \quad (7)$$

where  $\epsilon$  is energy and  $n_d(\epsilon)$  is the density of d-state.

The differential charge densities were calculated as follows:

$$\Delta\rho = \rho_{\text{AB}} - \rho_{\text{A}} - \rho_{\text{B}} \quad (8)$$

where  $\Delta\rho$  is differential charge density,  $\rho_{\text{AB}}$  is the charge density of system after reaction, the  $\rho_{\text{A}}$  is the charge density of the initial state of the system and the  $\rho_{\text{B}}$  is the charge density of the final state.

### 3. Results and discussion

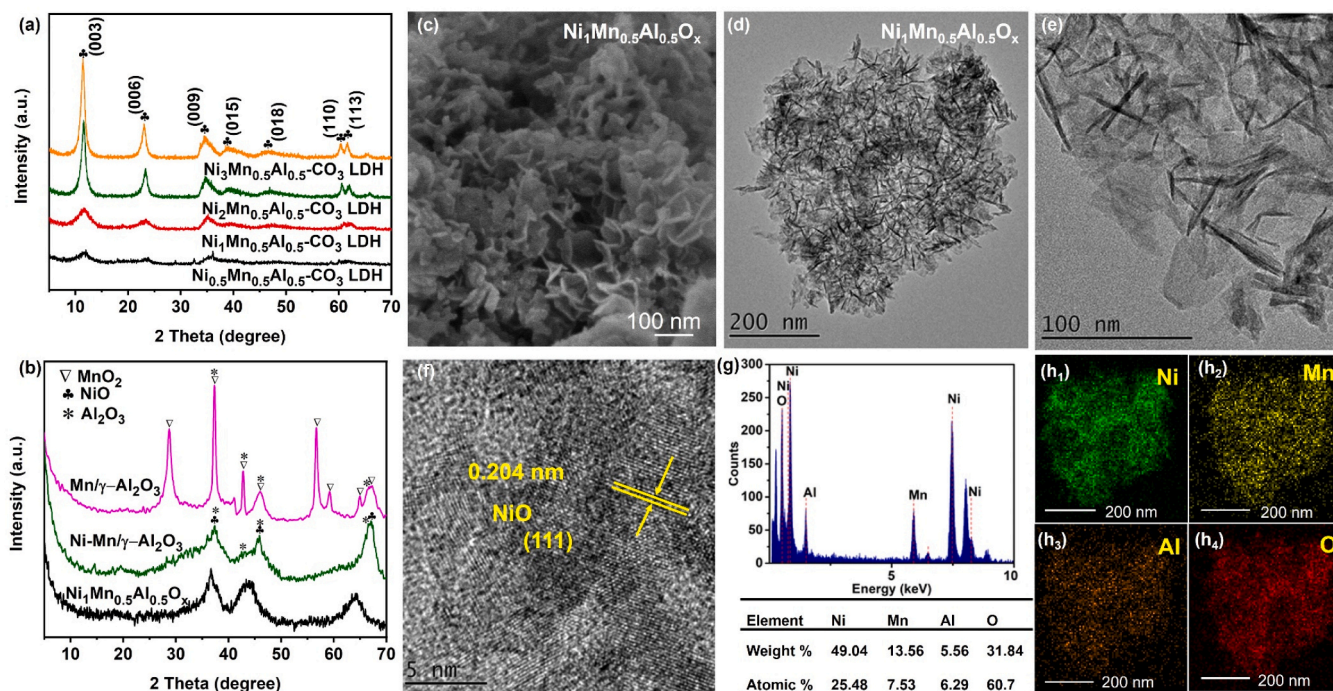
#### 3.1. Structural characterization

$\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{-CO}_3$  LDHs were first examined using XRD analyses (Fig. 1a).  $\text{NiMnAl-CO}_3$  LDHs samples all showed the typical layered structure similar to  $\text{MgAl}$  hydrotalcite in XRD patterns [37]. The peaks were observed at  $2\theta = 11.42^\circ$ ,  $23.04^\circ$ ,  $34.64^\circ$ ,  $38.9^\circ$ ,  $46.58^\circ$ ,  $60.4^\circ$  and  $61.64^\circ$ , corresponding to the reflections of the (003), (006), (009), (015), (018), (110), and (113) crystal planes, respectively. Fig. S1 showed that  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{-CO}_3$  LDHs were converted to metal oxide mixtures after calcination at 400 °C for 5 h. For  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{O}_x$  LDOs, there are three main diffraction peaks appearing at about  $37.25^\circ$ ,  $43.28^\circ$  and  $63.2^\circ$ , which belong to  $\text{NiO}$  phase (JCPDS No. 47-1049). Compared with  $\text{MnO}_2$  phase (JCPDS No. 24-0735) as the active component in  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  catalyst (Fig. 1b), no characteristic peaks of the  $\text{Mn}$  phase were observed in  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  and  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$ , but the characteristic peak of  $\text{NiO}$  phase (JCPDS No. 47-1049) appeared, suggesting that  $\text{Mn}$  species was highly dispersed after the introduction of  $\text{Ni}$  species. The morphology of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{-CO}_3$  LDH was also detected by FE-SEM (Fig. S2).  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{-CO}_3$  LDH displayed nanosheet morphology [38]. After calcination at 400 °C for 5 h, the morphology of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  LDO remained almost unchanged (Fig. 1c), similar to that of the LDH material, which was also confirmed in the HR-TEM images (Fig. 1d, e). The lattice fringe spacing of ca. 0.204 nm were observed, referring to the  $\text{NiO}$  (111) plane (Fig. 1f). In addition, TEM-EDS and elemental mapping analysis also confirmed the presence of  $\text{Mn}$  and  $\text{Ni}$  species in  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}$  in a highly dispersed state (Fig. 1g, h). There are similar conclusions that the introduction of another metal element can improve the dispersibility of  $\text{Mn}$  species, thereby improving the performance of the catalyst [12].

#### 3.2. Catalytic performance

The  $\text{DeNO}_x$  performance of  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{O}_x$  catalysts were investigated in a wide temperature range (100–300 °C). Fig. S3a showed the  $\text{DeNO}_x$  performance of  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{O}_x$  catalysts increased as reaction temperature increasing from 100 to 200 °C, and then started to decline slowly. In particular, the  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalyst exhibited the superior  $\text{NH}_3$ -SCR activity, especially at low temperatures (100–250 °C) with above 95%  $\text{NO}_x$  conversion. According to the results of the catalytic activity, if the  $y$  value exceeded 1, the  $\text{NO}_x$  conversion decreased, which

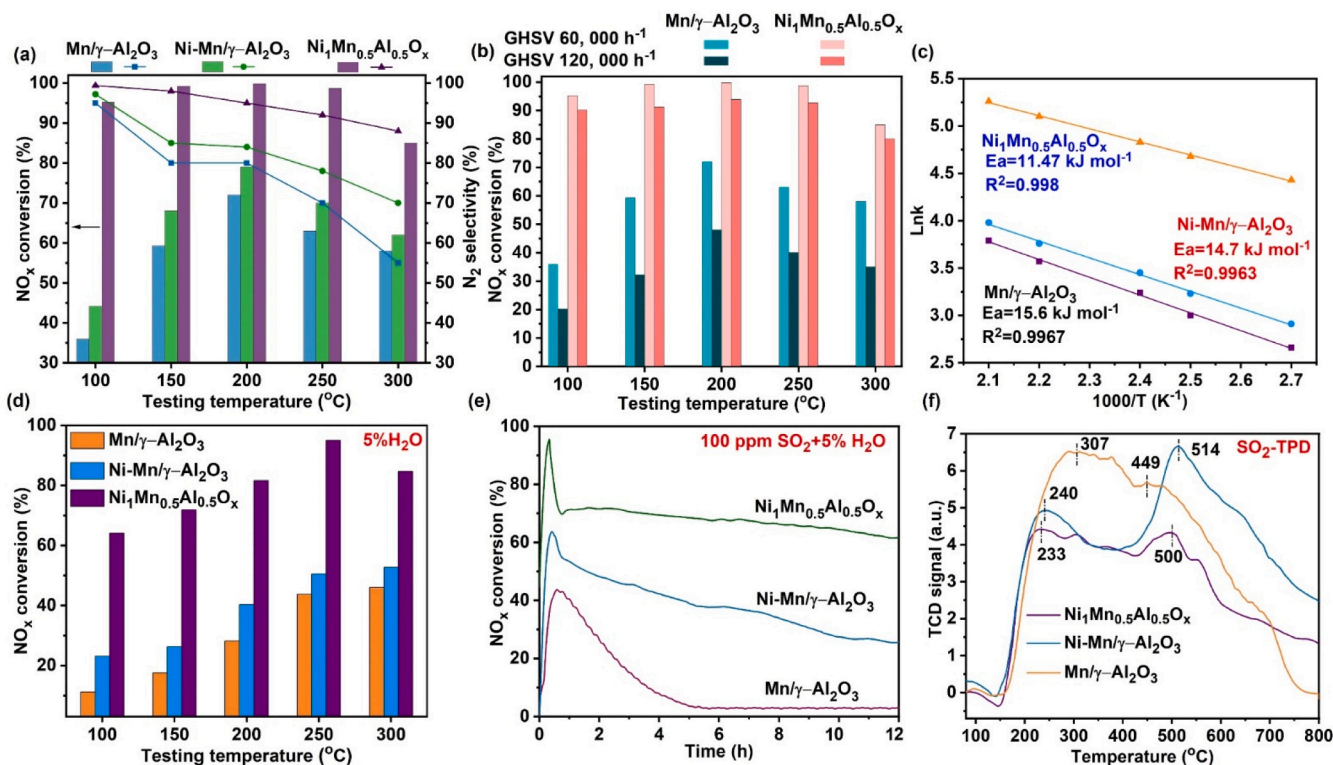




**Fig. 1.** XRD patterns of (a)  $\text{Ni}_y\text{Mn}_z\text{Al}_{1-z}\text{-CO}_3\text{LDH}$ s and (b)  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ ,  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalysts, (c) SEM image of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  LDO, (d-f) HR-TEM images of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$ , (g) TEM-EDS analysis of  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  solids, (h) EDS elemental mapping of Ni (h<sub>1</sub>), Mn (h<sub>2</sub>), Al (h<sub>3</sub>) and O (h<sub>4</sub>) in  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$ .

may be related to the maximum loading of the support [39,40]. Therefore,  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalyst exhibited the best performance at 100–300 °C, with the maximum  $\text{NO}_x$  conversion of nearly 100% at 200 °C. In contrast, the control  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$

exhibited poor low temperature activities, and the maximum  $\text{NO}_x$  conversions were only 72% and 79% at 200 °C, respectively (Fig. 2a). The  $\text{NH}_3$  conversion of catalysts was shown in Fig. S3b. At low temperatures, the  $\text{NH}_3$  conversion was very similar to that of the corresponding  $\text{NO}_x$



**Fig. 2.** (a)  $\text{NO}_x$  conversions and  $\text{N}_2$  selectivity, (b) The effect of different GHSVs (GHSV = 60,000  $\text{h}^{-1}$ , 120,000  $\text{h}^{-1}$ ), (c) Arrhenius plots of SCR reaction rates, (d) The effect of 5%  $\text{H}_2\text{O}$ , (e) The effect of 5%  $\text{H}_2\text{O}$  and 100 ppm  $\text{SO}_2$  co-existence, (f)  $\text{SO}_2$ -TPD profiles over  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ ,  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$ . Reaction conditions:  $[\text{NO}_x] = [\text{NH}_3] = 500 \text{ ppm}$ ,  $[\text{O}_2] = 5\%$ , balance Ar, total flow rate = 200 mL/min, GHSV = 60,000  $\text{h}^{-1}$ , catalyst 0.15 g.



conversion, suggesting that NO and NH<sub>3</sub> almost completely participate in the reaction. The distinct over consumption of NH<sub>3</sub> over Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> may be related to the NH<sub>3</sub> oxidation at high temperature (> 250 °C). Considering that some unexpected reactions may occurred during the SCR reaction, and one of the main defects of Mn-based catalysts is the poor N<sub>2</sub> selectivity. Fig. 2a showed the N<sub>2</sub> selectivity of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>1.5</sub>O<sub>x</sub>, Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, and Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. It can be seen that Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> displayed the lowest N<sub>2</sub> selectivity (~ 55%) with more N<sub>2</sub>O formation, but the N<sub>2</sub> selectivity of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> was greatly enhanced to more than 88% at 100–300 °C. Therefore, it indicated that both the N<sub>2</sub> selectivity and DeNO<sub>x</sub> activity of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> could be enhanced by the introduction of Ni. Moreover, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst still exhibited better DeNO<sub>x</sub> performance even at higher gas hourly space velocity (GHSV). Increasing the GHSV from 60,000 to 120,000 h<sup>-1</sup> resulted in only a slight decrease in NO<sub>x</sub> conversion for Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> at the entire test temperature (Fig. 2b). For instance, the NO<sub>x</sub> conversion of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> slightly decreased from 99.8% to 93% at 200 °C. In contrast, the NO<sub>x</sub> conversion of Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> decreased significantly from 72% to 48% under the same condition. The denitration performance of catalysts after calcination at 350–600 °C (Fig. S4). When the calcination temperature was increased from 350 °C to 600 °C, the DeNO<sub>x</sub> performance of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst remained at 89% at 200 °C, while Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> was only 24.6%. In addition, the optimum catalytic activity of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> prepared at a calcination temperature of 400 °C reached 99.8%. Therefore, the addition of Ni in combination with the LDHs-derived catalyst approach not only improved DeNO<sub>x</sub> activity and N<sub>2</sub> selectivity, but also broadened the operating temperature range of Mn-based catalysts.

As we all know, the reaction rate was closely related to the activation energy (E<sub>a</sub>). The lower the activation energy, the better the catalytic activity. The activation energy (E<sub>a</sub>) of low-medium temperature NH<sub>3</sub>-SCR reaction can be calculated from the slope of linear  $\ln k$  and 1000/T. In Fig. 2c, the Arrhenius curves of the three catalysts had a good linear relationship between  $\ln k$  and 1000/T. For the Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub>, the activation energy (11.47 kJ•mol<sup>-1</sup>) was lower than that of Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (14.7 kJ•mol<sup>-1</sup>) and Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (15.6 kJ•mol<sup>-1</sup>). The activation energy (E<sub>a</sub>) results further confirmed that the Ni modified Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst has higher catalytic activity in LT-SCR reaction.

### 3.3. SO<sub>2</sub> and/or H<sub>2</sub>O resistance of catalysts

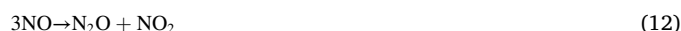
High resistance to SO<sub>2</sub> and H<sub>2</sub>O was quite necessary as they are usually present in the flue gas. SO<sub>2</sub> poisoning was mainly to form (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub>/NH<sub>4</sub>HSO<sub>4</sub> or metal sulfate species, which was even more serious for Mn-based catalysts. The effects of H<sub>2</sub>O are mainly attributed to compete with active sites for reactive species (NO<sub>x</sub>, NH<sub>3</sub>, etc.). Thus, we investigated the effect of SO<sub>2</sub> and/or H<sub>2</sub>O on the DeNO<sub>x</sub> activity of catalysts. With introducing 5% H<sub>2</sub>O, the DeNO<sub>x</sub> activity of Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> was lower than that of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (95%) and Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (50.5%) at 250 °C (Fig. 2d). In addition, the optimal reaction temperature of the catalyst shifted toward the high temperature from 200 to 250 °C, which indicated that H<sub>2</sub>O poisoning on the catalyst at low temperature was lower than that at high temperature. Fig. S5 showed the DeNO<sub>x</sub> activity of catalysts decreased after being exposed to 100 ppm SO<sub>2</sub> at 200 °C. After 6 h, the DeNO<sub>x</sub> activity of Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (~8.1%) was much lower than that of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (~73%) and Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (~45.2%).

In practical working conditions, both H<sub>2</sub>O and SO<sub>2</sub> coexisted in the flue gases. Thus, the effect of co-existence of 100 ppm SO<sub>2</sub> + 5% H<sub>2</sub>O on the DeNO<sub>x</sub> activity should be considered, and each experiment was performed for 12 h (Fig. 2e). The DeNO<sub>x</sub> activity dropped significantly from 72% to 3.1% (Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) and 79–25.6% (Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) at 200 °C, respectively. However, under the same condition, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> still achieved 61.7% NO<sub>x</sub> conversion. SO<sub>2</sub>-TPD experiments were implemented on catalysts (Fig. 2f). Two desorption peaks were obtained for all catalysts in the ranges of 235–337 °C and 448–507 °C, which

were the decomposition of (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub>/NH<sub>4</sub>HSO<sub>4</sub> and metal sulfate species, respectively. From the peak area of the desorption peak, Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> had more sulfate species deposited, so it was more susceptible to SO<sub>2</sub>, while Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst had less sulfate formation, and the negative effect of SO<sub>2</sub> on the DeNO<sub>x</sub> activity was weakened. Therefore, it was evident that the doping of Ni can alleviate the effect of H<sub>2</sub>O and/or SO<sub>2</sub> poisoning and improve the stability of the catalyst.

### 3.4. Transient kinetics studies for the formation of N<sub>2</sub>O

N<sub>2</sub>O and N<sub>2</sub> are often simultaneously formed in the NH<sub>3</sub>-SCR reaction of NO<sub>x</sub>. N<sub>2</sub>O may be formed by the following four reactions (Eqs. (9)–(12)):



In the NH<sub>3</sub>-SCR process, the NH<sub>3</sub> oxidation (Eq. (9)) was considered to be the main source of N<sub>2</sub>O formation [13]. Therefore, we investigated the catalytic performance of catalysts for the “NH<sub>3</sub> + O<sub>2</sub>” reaction. In Fig. 3a, the conversion of NH<sub>3</sub> became higher as the temperature increases. At 300 °C, the NH<sub>3</sub> conversion of Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> was 30%, which was slightly higher than Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (28%) and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (26.8%). More importantly, with the introduction of Ni, the formation of N<sub>2</sub>O can be reduced, especially at low temperatures. The amount of N<sub>2</sub>O produced by the “NH<sub>3</sub> + O<sub>2</sub>” reaction and the SCR reaction were also compared in this work. For Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst, the amount of N<sub>2</sub>O produced by the two reactions was almost equal (T = 250 °C, ~50 ppm N<sub>2</sub>O). For Ni-Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Mn/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, the amounts of N<sub>2</sub>O produced by “NH<sub>3</sub> + O<sub>2</sub>” reaction were much lower than the that by SCR reaction, which were 60.2 ppm, 67 ppm for “NH<sub>3</sub> + O<sub>2</sub>” reaction, and 82 ppm, 101 ppm for SCR reaction, respectively (Fig. 3a, b). Obviously, for the Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst, N<sub>2</sub>O was mainly generated by “NH<sub>3</sub> + O<sub>2</sub>” reaction, while the control catalysts have other pathways to produce N<sub>2</sub>O.

In order to further study other reaction pathways, we performed transient reaction studies on the above three samples. After the “NH<sub>3</sub> + O<sub>2</sub>” reaction reached a steady state, 500 ppm NO<sub>x</sub> was introduced in the flue stream, and the concentration changes of each component (N<sub>2</sub>O, NO, and NO<sub>2</sub>) were monitored. Fig. 3c displayed that the N<sub>2</sub>O concentration of the control catalysts increased significantly, while Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> was only slightly increased. Therefore, the non-selective catalytic “NH<sub>3</sub> + O<sub>2</sub> + NO<sub>x</sub>” reaction (Eqs. (10)–(11)) also contributed to the formation of N<sub>2</sub>O for the control catalysts, which was consistent with the conclusions of previous reports [13]. In addition, we also carried out NO oxidation reaction, as shown in Fig. 3d. There was nearly no N<sub>2</sub>O generation (~2.3 ppm N<sub>2</sub>O) for the three samples, indicating that N<sub>2</sub>O generation can ignore the NO<sub>x</sub> disproportionation reaction (Eq. (12)). For the control catalysts, “NH<sub>3</sub> + O<sub>2</sub>” and non-selective catalytic “NH<sub>3</sub> + O<sub>2</sub> + NO<sub>x</sub>” reactions were the main path of N<sub>2</sub>O production. The introduction of Ni could avoid the occurrence of non-selective catalytic “NH<sub>3</sub> + O<sub>2</sub> + NO<sub>x</sub>” reaction and reduce the amount of N<sub>2</sub>O production in SCR process. While “NH<sub>3</sub> + O<sub>2</sub>” reaction was the main reaction to produce N<sub>2</sub>O for Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub>, which showed the lower N<sub>2</sub>O production amount, occupying higher N<sub>2</sub> selectivity.

### 3.5. Surface physicochemical properties of catalysts

XPS analysis was performed to explore the chemical composition and surface element valence of catalysts (Fig. 4). In the O1s XPS spectra of three catalysts, two peaks due to the lattice oxygen (O<sub>b</sub>, 529.7–530 eV), and the surface adsorbed oxygen (O<sub>s</sub>, 531.3 eV) could be obtained. Due

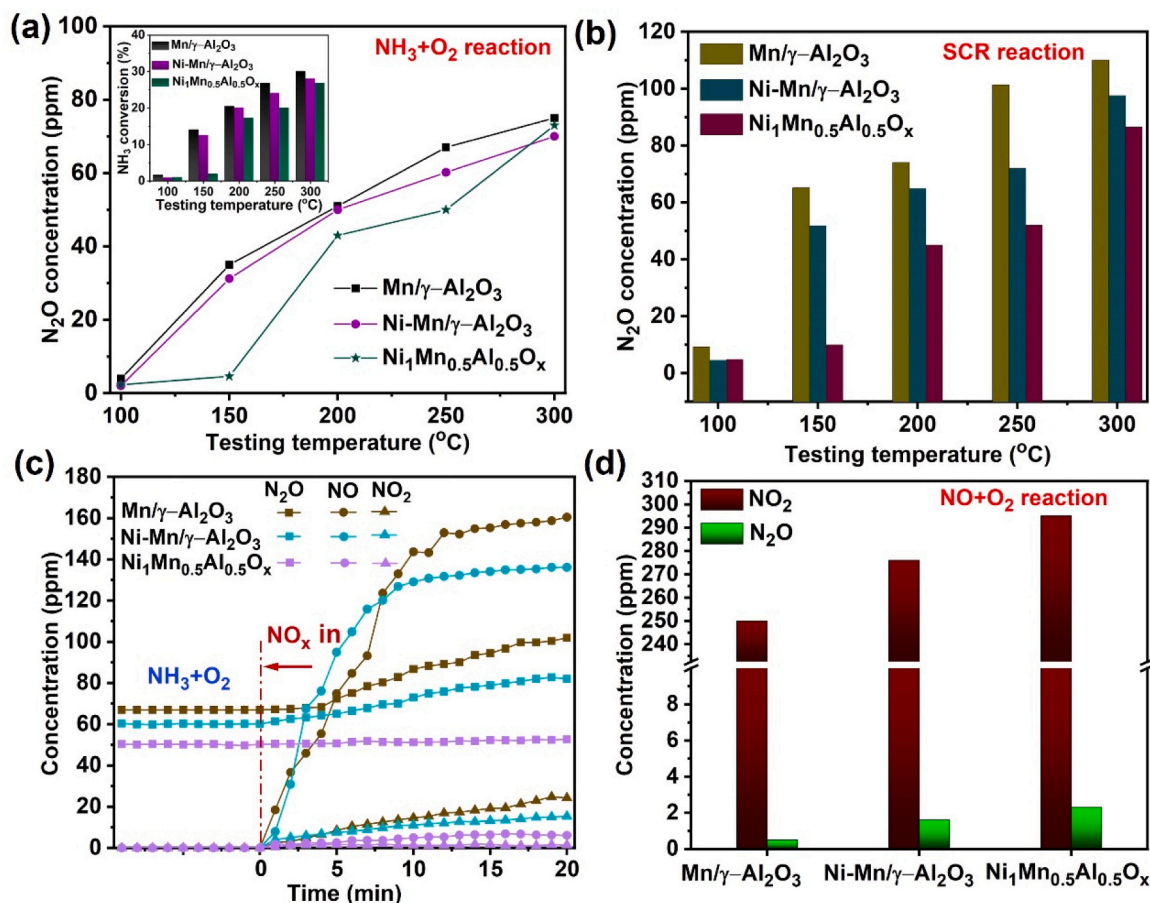


Fig. 3. (a) N<sub>2</sub>O concentration of NH<sub>3</sub> oxidation reaction and the inset figure is the NH<sub>3</sub> conversion on the catalysts, (b) N<sub>2</sub>O concentration of SCR reaction, (c) Transient reaction results, and (d) NO<sub>x</sub> oxidation reaction results over Mn/γ-Al<sub>2</sub>O<sub>3</sub>, Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub>. Reaction conditions: [NO<sub>x</sub>] = [NH<sub>3</sub>] = 500 ppm, [O<sub>2</sub>] = 5%, balance Ar, total flow rate = 200 mL/min, GHSV = 60,000 h<sup>-1</sup>, catalyst 0.15 g.

to the higher mobility, O<sub>α</sub> has higher activity in catalytic reactions than O<sub>β</sub> [41]. After the introduction of Ni, the O<sub>α</sub>/(O<sub>α</sub> + O<sub>β</sub>) atom-% fraction was increased following: Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (53.9%) > Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> (49.1%) > Mn/γ-Al<sub>2</sub>O<sub>3</sub> (43.6%). A higher proportion of O<sub>α</sub>/(O<sub>α</sub> + O<sub>β</sub>) favored the oxidation of NO, which led to the occurrence of “fast-SCR” reaction (4NH<sub>3</sub> + 2NO + 2NO<sub>2</sub> → 4 N<sub>2</sub> + 6 H<sub>2</sub>O).

The XPS analysis of Mn 2p showed that MnO<sub>x</sub> species existed in a mixed valence in all samples with two main peaks (Fig. 4b). For Mn 2p<sub>3/2</sub> peak, three peaks could be obtained, which were assigned to Mn<sup>4+</sup> (~643.3 eV), Mn<sup>3+</sup> (~641.6 eV), and Mn<sup>2+</sup> (~637.5 eV), respectively. For Mn-based catalysts, the catalytic cycle between Mn<sup>4+</sup> and Mn<sup>3+</sup> (MnO<sub>2</sub> → Mn<sub>2</sub>O<sub>3</sub> + O\*) was beneficial to the SCR reaction. Importantly, the total amount of Mn<sup>4+</sup> and Mn<sup>3+</sup> was increasing with the introduction of Ni. Since Mn<sup>4+</sup> was more susceptible to oxidation on Mn-based catalysts, it has an important role in catalytic reactions [32,42]. Obviously, the Mn<sup>4+</sup>/(Mn<sup>4+</sup> + Mn<sup>3+</sup> + Mn<sup>2+</sup>) atom-% fraction appeared to increase and followed the order of Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> (61.9%) > Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (51.2%) > Mn/γ-Al<sub>2</sub>O<sub>3</sub> (45.8%), and a moderate amount of Mn<sup>4+</sup> species can avoid excessive oxidation of NH<sub>3</sub>, thereby increasing N<sub>2</sub> selectivity.

In Fig. 4c, four peaks were mainly observed at 848–883 eV, which was due to Ni 2p<sub>3/2</sub> and Ni 2p<sub>1/2</sub> accompanied by two satellite peaks. Ni 2p<sub>3/2</sub> could be fitted into two peaks, corresponding to Ni<sup>3+</sup> (~855.6 eV) and Ni<sup>2+</sup> (~854.3 eV). For Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> and Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub>, the Ni<sup>2+</sup>/(Ni<sup>3+</sup> + Ni<sup>2+</sup>) atom-% fraction was 41.6% and 39.1%, respectively. Ni<sup>2+</sup> species were preferred for redox reactions and played an important role in improving SCR performance, which may be the electron transfer between Ni<sup>2+</sup> and Mn<sup>4+</sup> in a process similar to Mn<sup>4+</sup> +

Ni<sup>2+</sup> ↔ Mn<sup>3+</sup> + Ni<sup>3+</sup> [43]. Therefore, the relatively high contents of Mn<sup>4+</sup>, Ni<sup>2+</sup> and O<sub>α</sub> in Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> promoted its excellent catalytic activity. According to ICP-MS test and XPS analysis, the Mn and Mn<sup>4+</sup> contents were obtained. In Fig. 4d, the Mn content of Mn/γ-Al<sub>2</sub>O<sub>3</sub>, Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalysts was 1.8%, 0.9% and 14.6% respectively, which was similar to the dosage of preparation. In addition, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> has more highly dispersed Mn<sup>4+</sup> species, which was beneficial to LT-SCR reaction. Compared with Mn/γ-Al<sub>2</sub>O<sub>3</sub> (135.5 m<sup>2</sup>/g) and Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> (186.28 m<sup>2</sup>/g), the introduction of Ni could effectively increase the specific surface area of the sample (Table S1). In addition, the specific surface area of LDOs obtained by high temperature calcination of LDHs became larger. Especially, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> possessed the largest specific surface area (374.15 m<sup>2</sup>/g), which facilitated the adsorption of NO<sub>x</sub> and NH<sub>3</sub> reactants.

In order to study the effect of Ni doping on the reducibility of the catalyst, H<sub>2</sub>-TPR experiments of the catalyst were performed (Fig. 5a). For the control Mn/γ-Al<sub>2</sub>O<sub>3</sub> sample, two H<sub>2</sub>-TPR peaks were around ~263 and 567 °C, which may be attributed to the following successive reduction steps: MnO<sub>2</sub> → Mn<sub>2</sub>O<sub>3</sub> → Mn<sub>3</sub>O<sub>4</sub> → MnO [22,44]. For the Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> and Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub>, two broad peaks were still observed. Compared with the H<sub>2</sub>-TPR curve of Mn/γ-Al<sub>2</sub>O<sub>3</sub>, the temperature of the first reduction peak of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> shifted to the lower temperature. The second reduction peak at high temperature (567–644 °C) may be due to the superposition of the reduction of Ni<sup>3+</sup> → Ni<sup>2+</sup> → Ni<sup>0</sup> and MnO<sub>x</sub> species. In addition, the H<sub>2</sub>-TPR peak areas of Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> were larger than that of Mn/γ-Al<sub>2</sub>O<sub>3</sub>. The H<sub>2</sub> consumptions of samples were quantified through a pulse experiment,

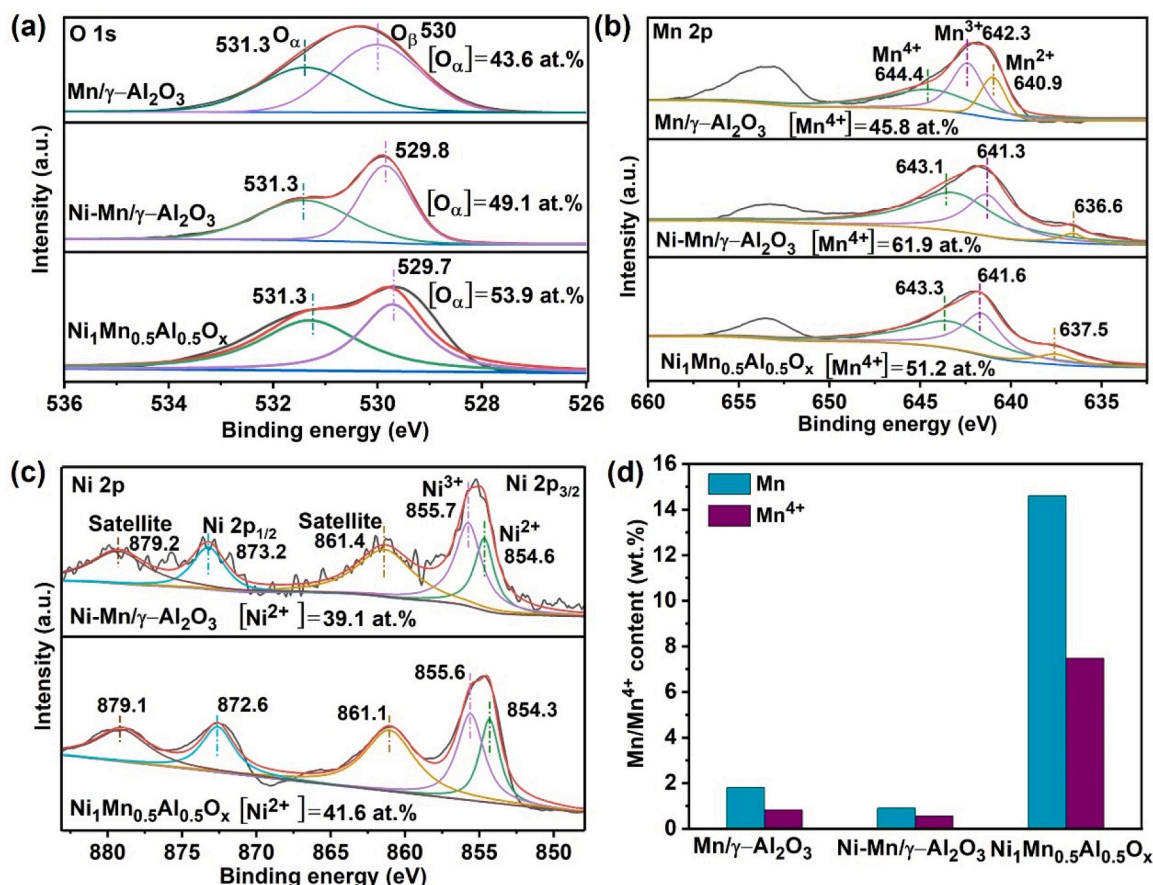


Fig. 4. XPS results of (a) O 1s, (b) Mn 2p, and (c) Ni 2p over catalysts; (d) Mn/Mn<sup>4+</sup> content of catalysts.

which followed the sequence of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (2.15 mmol/g) > Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> (0.45 mmol/g) > Mn/γ-Al<sub>2</sub>O<sub>3</sub> (0.18 mmol/g). The improvement may be attributed to the synergistic influence between Ni and Mn species. Therefore, combined with the reduction peak temperature and H<sub>2</sub> consumption, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> prepared by LDH calcination has higher Mn and Cu species reduction ability, which means that the surface oxygen mobility increased and the SCR reaction proceeded rapidly.

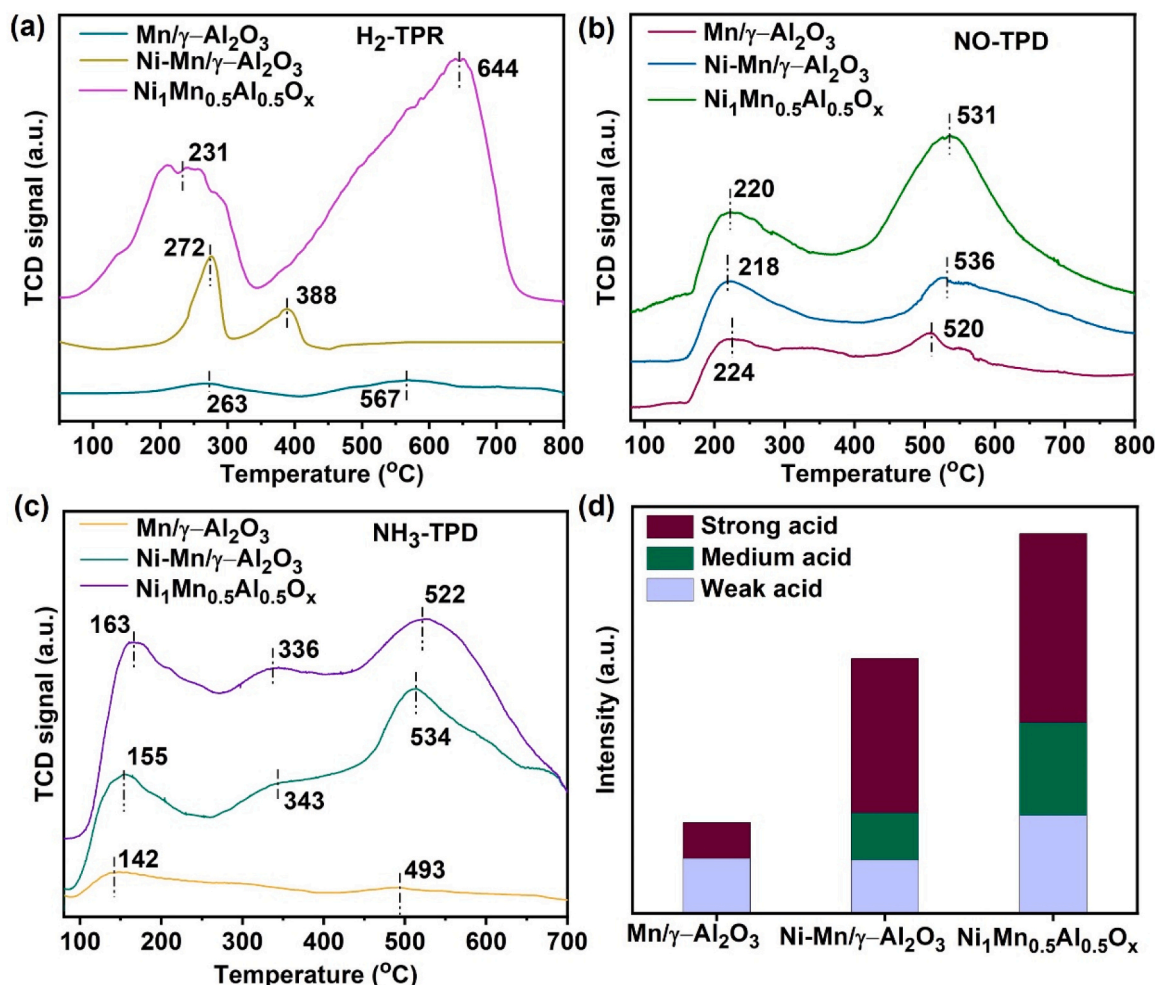
It was well known that the reaction between adsorbed NH<sub>3</sub> and adsorbed NO species was the main reaction at low temperature (L-H mechanism). So, the NO-TPD analysis of all samples were shown in Fig. 5b, and all samples showed two desorption peaks. The peaks at low temperature (~200 °C) could be attributed to the desorption of monodentate nitrate species, while the peaks at high temperature (~500 °C) were the desorption of bidentate/bridged nitrate species [4,45]. Notably, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> showed the largest desorption peak area, especially at low temperatures, indicating that it had the highest adsorption capacity for NO. In addition, NH<sub>3</sub>-TPD analysis of all samples were studied (Fig. 5c). For the control Mn/γ-Al<sub>2</sub>O<sub>3</sub> sample, two desorption peaks were displayed at ~142 and 493 °C, assigned to weak and strong acid sites, respectively. With the introduction of Ni species, a distinct new desorption peak appeared at ~336 °C, corresponding to medium acid sites. The amounts of desorbed NH<sub>3</sub> (surface acidity) over Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub>, Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub>, and Mn/γ-Al<sub>2</sub>O<sub>3</sub> were compared by integrating the NH<sub>3</sub>-TPD curves (Fig. 5d), which followed the order of Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> (240.5 μmol/g) > Ni-Mn/γ-Al<sub>2</sub>O<sub>3</sub> (137.2 μmol/g) > Mn/γ-Al<sub>2</sub>O<sub>3</sub> (105.1 μmol/g). Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst was visible larger than that of the control samples, which suggested that the addition of Ni species increased the amount of acid sites, and significantly improved the adsorption capacity of NH<sub>3</sub>.

### 3.6. In situ DRIFTS studies

After exposure to 1000 ppm NH<sub>3</sub>/N<sub>2</sub> at 200 °C for different times, the adsorption of NH<sub>3</sub> on Mn/γ-Al<sub>2</sub>O<sub>3</sub> and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> were studied by in-situ DRIFTS (Fig. 6a, b). The characteristic peaks of amide (–NH<sub>2</sub>) species (1556, and 1531 cm<sup>–1</sup>) [46,47], NH<sub>4</sub><sup>+</sup> (1690, and 1694 cm<sup>–1</sup>) bonded to Brønsted acid sites [46], and coordinated NH<sub>3</sub> (1617, 1622, 1177, and 1261 cm<sup>–1</sup>) linked to Lewis acid sites appeared with time increase [48]. Besides, the peak corresponding to the –NH species (1461 cm<sup>–1</sup>) appeared in the DRIFTS spectra of NH<sub>3</sub> adsorption over Mn/γ-Al<sub>2</sub>O<sub>3</sub> catalyst, which was caused by excessive hydrogen absorption of NH<sub>3</sub>, but can hardly be found in the spectrum of catalyst Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> [46]. Liu et al. [49] and Kapteijn et al. [15] reported that the –NH species were unable to react with NO to form N<sub>2</sub> instead of N<sub>2</sub>O species. The peak intensity of adsorbed NH<sub>3</sub> species indicated that the introduction of Ni could improve the adsorption behavior of NH<sub>3</sub> on the catalyst surface. The adsorption and activation of NH<sub>3</sub> could be promoted by the synergistic effect due to Ni doping, which was consistent with the results of NH<sub>3</sub>-TPD (Fig. 5c).

Fig. 6c, d showed the in-situ DRIFT spectra of Mn/γ-Al<sub>2</sub>O<sub>3</sub> and Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalysts exposed to 2000 ppm NO + 5% O<sub>2</sub>/Ar gas flow. The adsorbed NO<sub>2</sub> species (1612, 1622 cm<sup>–1</sup>) [50], bidentate nitrate (1551–1582 cm<sup>–1</sup>) [51,52], monodentate nitrate species (1254, 1277, 1310 cm<sup>–1</sup>) were observed [53], where the peaks of chelating nitrite and bridging nitrate overlapped [52]. The bidentate nitrate and monodentate nitrate increased with increasing temperature. Comparing the peak intensity of NO<sub>x</sub> species, Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> was stronger than Mn/γ-Al<sub>2</sub>O<sub>3</sub>, indicating that Ni and Mn have a better ability to adsorb and activate NO<sub>x</sub>. Therefore, it was evident that the doping of Ni enriched the active sites for adsorption and activation of reactive species, and accelerated the NH<sub>3</sub>-SCR process on the surface of





**Fig. 5.** (a)  $\text{H}_2$ -TPR profiles, (b) NO-TPD profiles, (c)  $\text{NH}_3$ -TPD profiles, (d) surface acidity distribution over different catalysts; In situ DRIFTS spectra obtained after a 30 min exposure to 1000 ppm  $\text{NH}_3$  over (e)  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ , (f)  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalysts at 200  $^\circ\text{C}$ ; In situ DRIFTS spectra obtained after a 30 min exposure to 2000 ppm  $\text{NO} + 5\% \text{O}_2$  over (g)  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (h)  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalysts at 200  $^\circ\text{C}$ .

$\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalyst.

In-situ DRIFTS study of  $\text{NO} + \text{O}_2$  reaction with pre-adsorbed  $\text{NH}_3$  was recorded at 200  $^\circ\text{C}$ , as illustrated in Fig. 7. After introducing  $\text{NO} + \text{O}_2$  in the reaction cell, these bands assigned to the adsorbed  $\text{NH}_3$  species gradually weaken and disappeared, while those peaks ascribing to the adsorbed  $\text{NO}_x$  species of  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (1307, 1540, 1559, 1585, and 1614  $\text{cm}^{-1}$ ) and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  (1252, 1277, 1544, 1573, and 1617  $\text{cm}^{-1}$ ) gradually emerged with time (Fig. 7a, b) [54]. The difference was that more  $\text{NH}_3$  species adsorbed on  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  was consumed than  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ . The above results revealed that the reaction of gaseous  $\text{NO} + \text{O}_2$  and pre-adsorbed  $\text{NH}_3$  can proceed through the E-R mechanism on the both catalysts. In comparison,  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  presented a better activating ability of  $\text{NH}_3$  than  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ , which could accelerate more active  $\text{NH}_3$  species to react with  $\text{NO}_x$ , thereby enhancing the  $\text{NH}_3$ -SCR performance.

The reactivity of  $\text{NH}_3$  with pre-adsorbed  $\text{NO} + \text{O}_2$  was investigated at 200  $^\circ\text{C}$  by DRIFTS analysis, as presented in Fig. 7c, d. After pretreatment by  $\text{NO} + \text{O}_2$ , various bands assigned to adsorbed  $\text{NO}_x$  species could be found both in the spectra of  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$ . The bands of adsorbed  $\text{NO}_x$  species quickly decreased in intensity and all bands nearly vanished when  $\text{NH}_3$  was added in for 10 min. Additionally, the bands ascribing to adsorbed  $\text{NH}_3$  species of  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (1311, 1534, 1543, 1559, 1565, and 1609  $\text{cm}^{-1}$ ) and  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  (1184, 1252, 1278, 1520 and 1610  $\text{cm}^{-1}$ ) arose after 2 min [55]. The above observations suggested that the reaction of  $\text{NH}_3$  and pre-adsorbed  $\text{NO}_x$  species could occur on these two catalysts via L-H mechanism. Combined with

DRIFTS adsorption and  $\text{NH}_3/\text{NO}$ -TPD result (Fig. 5), the Ni doping enhanced the total amount of  $\text{NH}_3$  and  $\text{NO}_x$  adsorbed on  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$ , which can accelerate the SCR process via the L-H and E-R mechanisms.

### 3.7. DFT calculations

DFT was used to calculate in depth the properties of the catalyst surface. The charge density difference plots for  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{Ni-Mn}/\gamma\text{-Al}_2\text{O}_3$  catalysts were employed to detect the catalysts surface. In Fig. 8a, b, oxygen accepted electrons from Mn in  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  catalyst, implicating that Mn atoms and oxygen atoms were connected by ionic bonds. For  $\text{Mn}_{5c}$  site, larger cyan areas appeared at its center and top, belonging to its un-occupation orbit. The un-occupation orbit could accommodate additional electrons, which can be responsible for Lewis acid site [56], thus,  $\text{Mn}_{5c}$  showed strong Lewis acidity. The charge density difference plots for Ni doping of  $\text{NiMnAlO}_x$  catalyst was shown in Fig. 8c, d. The apparent electrons transfer could be found between Mn atom and oxygen atom. Electrons were also observed accumulating between  $\text{Mn}_{3c}$  and oxygen atom, and a few electrons accumulated at the center of  $\text{Mn}_{3c}$ . In addition,

$\text{Mn}_{3c}$  was surrounded by a flower-like electron loss region with a small amount of unoccupied orbitals present, contributing to the generation of moderate Lewis acids in the  $\text{NiMnAlO}_x$  catalyst, and this result corroborated the result of TPD experiment. Strong electron transition could be found between Mn atom and oxygen atom, or Ni atom and

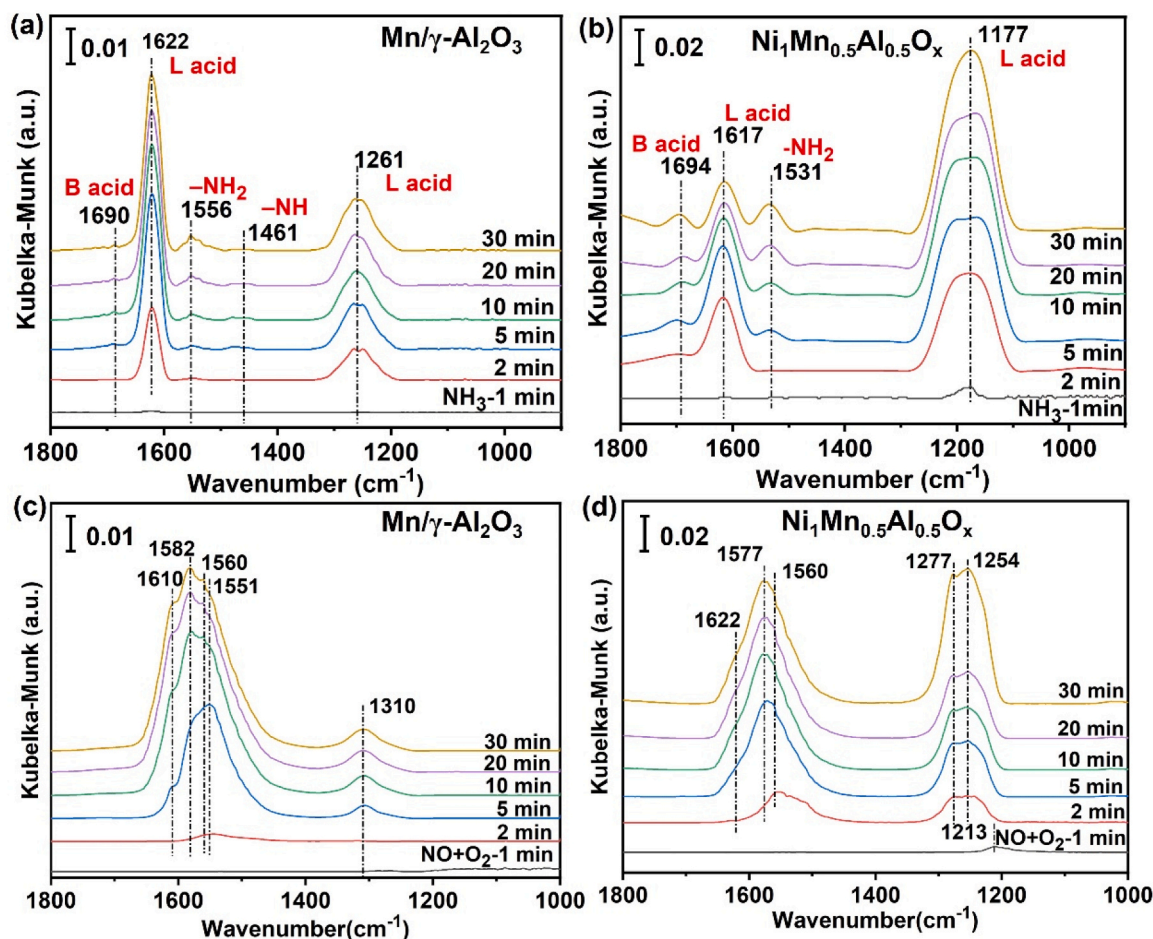


Fig. 6. In situ DRIFTS spectra obtained after a 30 min exposure to 1000 ppm  $\text{NH}_3$  over (a)  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ , (b)  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalysts at  $200^\circ\text{C}$ ; In situ DRIFTS spectra obtained after a 30 min exposure to 2000 ppm  $\text{NO} + 5\% \text{O}_2$  over (c)  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (d)  $\text{Ni}_1\text{Mn}_{0.5}\text{Al}_{0.5}\text{O}_x$  catalysts at  $200^\circ\text{C}$ .

oxygen atom on  $\text{NiMnAlO}_x$  surface, which may lead to change in valence of Mn and Ni ions through the process of  $\text{Mn}^{4+} + \text{Ni}^{2+} \leftrightarrow \text{Mn}^{3+} + \text{Ni}^{3+}$ .

PDOS analysis of the d-band of Mn showed that the d-band of Mn of  $\text{NiMnAlO}_x$  was more localized and had a greater degree of spin polarization compared to the d-band of  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (Fig. 8e). The d-band of  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  had a low bandwidth of spins broadly distributed between  $-6$  and  $3$  eV, whereas the d-band of  $\text{NiMnAlO}_x$  had a localized and narrow peak near  $-5$  to  $-2$  eV, so that more active electrons were available for the Mn site in  $\text{NiMnAlO}_x$  oxides to participate in the interaction. The d up spin of  $\text{NiMnAlO}_x$  had a small peak only at the Fermi level for a small unoccupied orbital, indicating that the d upward spin was weakly acidic, in agreement with the charge density difference results. The d-band down spins of both  $\text{NiMnAlO}_x$  and  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  existed unoccupied orbitals, suggesting that both of their down spins exhibited electron-acquiring ability and acidity. But there were fewer occupied state orbitals in the d-band down spins of  $\text{NiMnAlO}_x$ . Interestingly, these phenomena were also reflected in the coordination field effects of Mn (Fig. 8f-h). The d-band suborbital energy levels of Mn split in  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  showed a high degree of degeneracy than in  $\text{NiMnAlO}_x$ . The energy gap between the  $t_{2g}$  and  $e_g$  orbitals in the up and down spins of the d-band of Mn in the  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  system was about  $0.1$  eV and  $0.4$  eV, respectively, indicating that there was an obvious Jahn-Teller effect for the Mn element in  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$ . In contrast, the  $t_{2g}$  and  $e_g$  orbitals in the up and down spins of the d-band of Mn in  $\text{NiMnAlO}_x$  were highly degenerated, leading to the disappearance of the energy gap between the  $t_{2g}$  and  $e_g$  orbitals, and this Jahn-Teller degradation explained the localized and intensified d-band of Mn in  $\text{NiMnAlO}_x$  [37].

The  $E_{\text{ads}}$  of  $\text{NH}_3$  and  $\text{NO}$  on the  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (110) and  $\text{NiMnAlO}_x$

(111) surface were evaluated (Fig. 9). The result showed that the adsorption reactions of  $\text{NH}_3$  molecular on  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  and  $\text{NiMnAlO}_x$  catalysts were exothermic reaction, with  $E_{\text{ads}}$  of  $-1.27$  eV and  $-1.17$  eV and bond lengths of  $2.05$  Å and  $2.20$  Å, respectively, suggesting that  $\text{NH}_3$  and  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  catalyst had strong bonding ability. The Mn sites served as Lewis acid site in these two configurations, and the lower adsorption energy indicated that Ni specie may weaken the strong surface Lewis acidity to moderate surface Lewis acid of  $\text{NiMnAlO}_x$  catalyst. According to the literature and the result of our transient kinetics studies, the  $\text{N}_2\text{O}$  formation in  $\text{NH}_3$ -SCR reaction was mainly through  $\text{NH}_3$  oxidation pathway [57]. More moderate Lewis acidity of Mn active site may lead to less  $\text{N}_2\text{O}$  formation via  $\text{NH}_3$  oxidation pathway.

The  $E_{\text{ads}}$  of  $\text{NO}$  on the surface Mn site enhanced from  $0.22$  eV to  $-0.78$  eV, and bond length decreased from  $2.36$  Å to  $1.76$  Å on the  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (110) and  $\text{NiMnAlO}_x$  (111), respectively. The higher  $E_{\text{ads}}$  for  $\text{NO}$  on the  $\text{NiMnAlO}_x$  (111) indicated that the Ni species greatly enhanced the interaction between  $\text{NO}$  and Mn site, well supporting the view that Ni increased the active sites for adsorption and activation, as mentioned in DRIFT and NO-TPD experiments. For absolute value of adsorption energy was more than  $-0.3$  eV,  $\text{NH}_3$  and  $\text{NO}$  molecular can be chemisorbed over the surface of  $\text{NiMnAlO}_x$  (111), while  $\text{NO}$  cannot be adsorbed on the  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (110). Comparing the adsorption energy between  $\text{NO}$  and  $\text{NH}_3$ ,  $\text{NH}_3$  adsorption energy was higher on  $\text{Mn}/\gamma\text{-Al}_2\text{O}_3$  (110) surface but lower on  $\text{NiMnAlO}_x$  surface, which indicated that  $\text{NH}_3$ -SCR reaction preferred L-H reaction route while doping Ni on  $\text{NiMnAlO}_x$  LDO. Moreover,  $\text{NH}_3$ -SCR on catalyst at low temperature followed L-H mechanism more effectively.

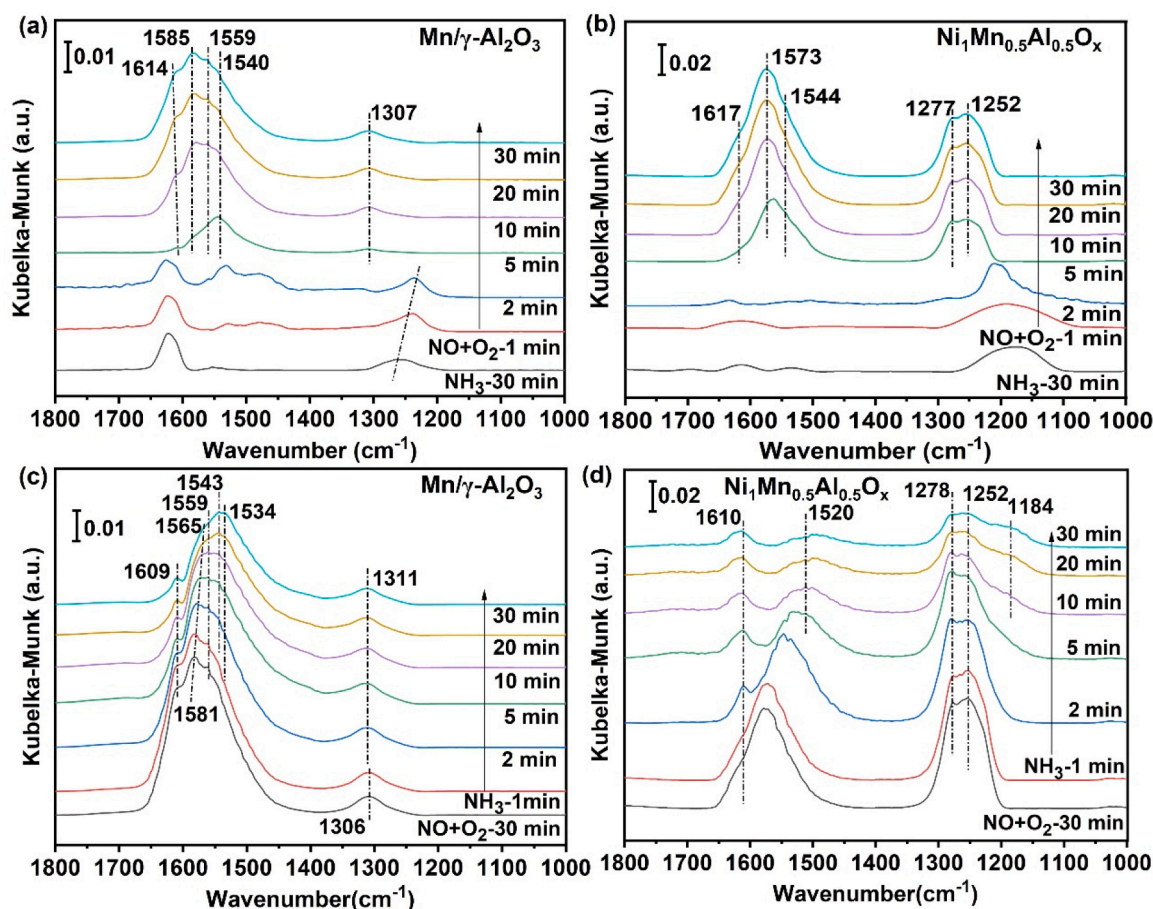


Fig. 7. In situ DRIFTS spectra of the reaction between 2000 ppm NO + 5% O<sub>2</sub> and pre-adsorbed NH<sub>3</sub> over (a) Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (b) Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> and 1000 ppm NH<sub>3</sub> reacting with pre-adsorbed NO + O<sub>2</sub> over (c) Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (d) Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> at 200 °C.

Interestingly, we found that the adsorption of small molecular on the NiMnAlO<sub>x</sub> catalyst was accompanied by a magnetic moment reversal of the active center and the catalyst as a whole, which permitted the Mn active center to interact with the small molecule in an electronic state more favorable for small molecule reactions [58]. In contrast to the ferromagnetism of system Mn/γ-Al<sub>2</sub>O<sub>3</sub>, the anti-ferromagnetism of system NiMnAlO<sub>x</sub> allowed the doping Mn active sites therein to switch between opposite magnetic moments. The Ni site in the anti-ferromagnetic bulk of NiO had opposite magnetic moments of  $-1.82$  and  $1.82 \mu_B$  (Fig. 10a). The antiferromagnetic cell of NiO allowed for opposite magnetic moments between two adjacent layers of Ni atoms in its (111) surface (Fig. 10b), with the surface Ni atomic layer having a magnetic moment at  $\sim 1.4 \mu_B$  relative to the deeper Ni. When Mn was doped into the superficial atomic Ni layer on the surface of NiO (111), Mn and the co-layered Ni atoms exhibited opposite magnetic moments due to the large magnetic moment ( $4.32 \mu_B$ ) of Mn (Fig. 10c), which made the magnetic moment of the Mn site easily reversed (Fig. 10d). The reversal of the magnetic moments of the Mn sites led to the reversal of the magnetic moments of the same layer of Ni and the adjacent layer of Ni, which will lead to the reversal of the magnetic moment of the catalyst as a whole. This reversal of magnetic moment occurred when the catalyst adsorbed paramagnetic NO (Fig. 10f), which made the binding of NO to Mn site stronger, as evidenced by a significant increase in the adsorption energy of NO at Mn site. The binding of NO to Mn site weakened the huge magnetic moment of Mn site, which led to the stability of the overall antiferromagnetic state of the catalyst. In contrast, the interaction of the non-magnetic NH<sub>3</sub> with the Mn site did not have this magnetic moment reversal and did not cause the magnetic moment of the Mn site weakened (Fig. 10e), which also prevented the

over-bonding of NH<sub>3</sub> and Mn. In ferromagnetic Mn/γ-Al<sub>2</sub>O<sub>3</sub>, the interaction of the Mn site with NH<sub>3</sub> slightly weakened the magnetism of the adsorbed site, which may be due to over-bonding of NH<sub>3</sub> to the Mn site. The weak interaction of NO with Mn of Mn/γ-Al<sub>2</sub>O<sub>3</sub> did not change the magnetic moment of Mn, while the interaction caused NO to take on the up magnetic moment in ferromagnetic system.

In order to gain insight into the mechanism and influence of the degradation of the Jahn-Teller effect and magnetic moment reversal of Mn sites on the surface adsorption process, the partial density of state (PDOS) for NO/NH<sub>3</sub> adsorption models of Mn/γ-Al<sub>2</sub>O<sub>3</sub> and NiMnAlO<sub>x</sub> were investigated (Fig. 11). For NO adsorption, the  $\pi^*$  orbital resonated only slightly with the broad d-band up spin of Mn/γ-Al<sub>2</sub>O<sub>3</sub> about the Fermi level (Fig. 11a). While for NO adsorbed on Mn site of NiMnAlO<sub>x</sub> surface, not only the  $\pi^*$  orbital interacted with both up spin and down spin of the localized d band, but also the  $\pi$  orbital involved in the bonding with down spin of d band, which was indicated by the  $\pi$ -d bonding and  $(\pi-d)^*$  bonding at  $-6.6$  eV and  $-2.7$  eV, respectively. This bonding mode was due to the reversal of the magnetic moment of the Mn site in NiMnAlO<sub>x</sub> upon NO adsorption, which caused the d-band up spin to become a state with a large number of empty orbitals, while the electron-accessing ability of the empty orbitals was further strengthened by the effect of the Jahn-Teller degeneracy, thus permitting single electrons on the  $\pi^*$  orbitals in NO to enter into empty orbitals on the d-band up spin for interactions. Similarly, the down spin of Mn d band after the magnetic moment reversal had active electrons that could be injected into the down spin empty orbitals of  $\pi^*$  of NO and could interact with the  $\pi$  orbitals to form a series of bonding and antibonding orbitals. Consequently, the multiple bonding between NO molecule and NiMnAlO<sub>x</sub> markedly enhanced the NO chemisorption. Fig. 11c,



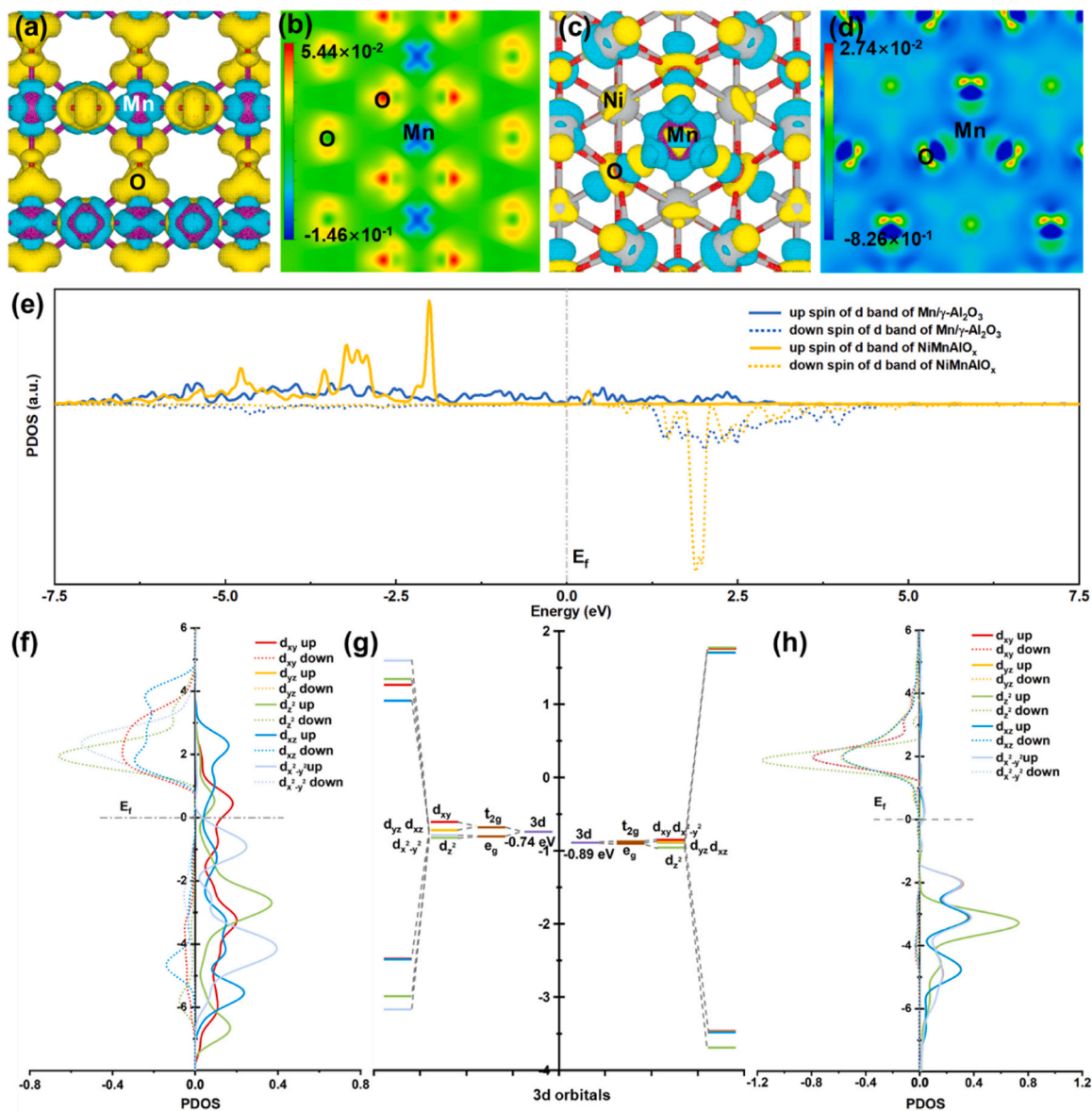


Fig. 8. The charge density difference plots and 2D display of (a, b) Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (c, d) NiMnAlO<sub>x</sub>. Cyan area represents loss of electron, yellow area represents gain of electron. (e) PDOS profiles of the Mn site of Mn/γ-Al<sub>2</sub>O<sub>3</sub> and NiMnAlO<sub>x</sub>, 3d orbital-resolved PDOS profiles for Mn site on (f) clean Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (h) NiMnAlO<sub>x</sub>, (g) schematic band structure of the Mn site, the left and right side of the vertical axis belong to Mn/γ-Al<sub>2</sub>O<sub>3</sub> and sulfated NiMnAlO<sub>x</sub>, respectively.

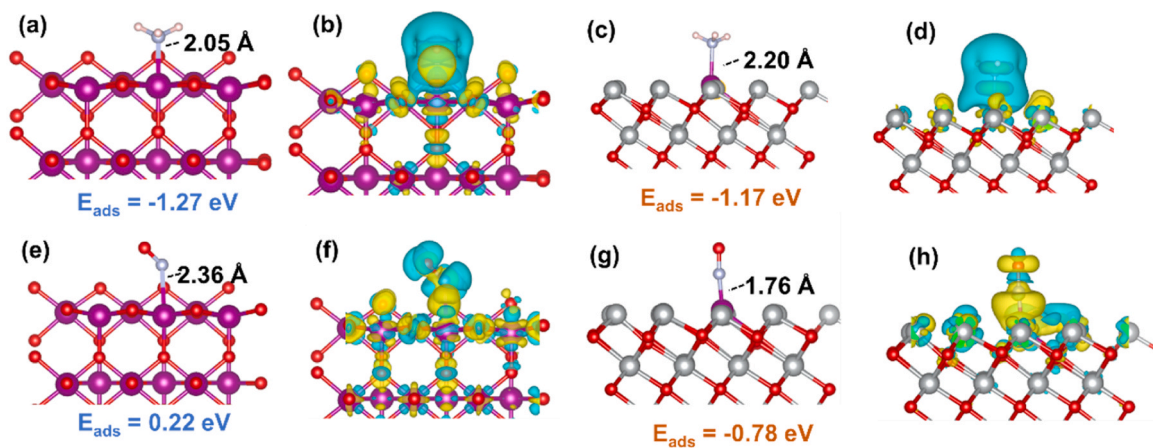
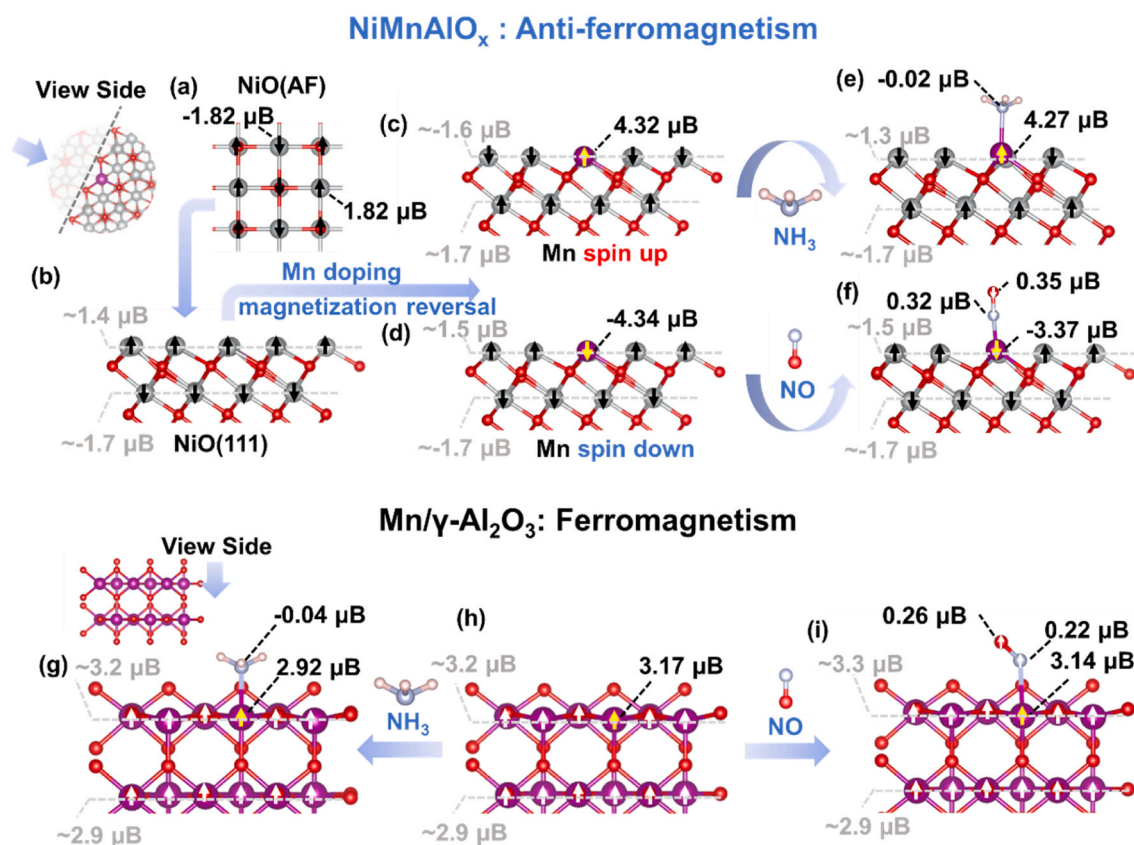


Fig. 9. Optimized configurations and corresponding charge density differences of NO and NH<sub>3</sub> adsorbed on (a, b and e, f) Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (c, d and g, h) NiMnAlO<sub>x</sub>.



**Fig. 10.** Schematic diagram of the magnetization reversal process during small molecular adsorption on Mn/γ-Al<sub>2</sub>O<sub>3</sub> and NiMnAlO<sub>x</sub> and the magnetic moment diagram of (a) NiO bulk, (b) NiO (111), (c) clean and (e) NH<sub>3</sub> adsorption on NiMnAlO<sub>x</sub> with up-magnetic moment Mn site, (d) clean and (f) NO adsorbed on NiMnAlO<sub>x</sub> with down-magnetic moment Mn site, and (h) clean, (g) NO, and (i) NH<sub>3</sub> adsorbed on Mn/γ-Al<sub>2</sub>O<sub>3</sub>.

d compared the bonding state of NH<sub>3</sub> molecule adsorbed on Mn/γ-Al<sub>2</sub>O<sub>3</sub> and NiMnAlO<sub>x</sub> surfaces. The n orbitals of NH<sub>3</sub> shifted to the low energy level after bonding to the Mn site of Mn/γ-Al<sub>2</sub>O<sub>3</sub>, with the broaden n-d bonding and (n-d)\* anti-bonding evolved in up spin. Analogously, the d band of Mn site of NiMnAlO<sub>x</sub> surfaces exhibited an obvious interaction with the n orbitals of NH<sub>3</sub>. However, the Jahn-Teller degradation led to the localization of the up spin on the d band and the reduction of the empty orbitals, so that the (n-d)\* anti-bonding orbitals were below the Fermi level, suggesting the filling of electrons, and simultaneously the reduction of resonance of the down spin n and d bands, which reduced the stability of the incorporation between NH<sub>3</sub> and NiMnAlO<sub>x</sub>. In addition, compared to Mn/γ-Al<sub>2</sub>O<sub>3</sub>, the d band of the Mn atom of NiMnAlO<sub>x</sub> shifted to higher energy level after combining with NH<sub>3</sub>. These also further explained the lower adsorption energy for NH<sub>3</sub> on NiMnAlO<sub>x</sub> than on Mn/γ-Al<sub>2</sub>O<sub>3</sub>.

#### 4. Conclusions

In this work, the overall catalytic performance of LDHs-derived NiMnAlO<sub>x</sub> were studied in the NH<sub>3</sub>-SCR reaction of NO<sub>x</sub>. The preferred Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> exhibited excellent LT-SCR catalytic activity with a NO<sub>x</sub> conversion of exceeded 95% at 100–250 °C. In the transient reaction, we demonstrated that the Ni-modified Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst could effectively inhibit the non-selectively catalytic "NH<sub>3</sub> + O<sub>2</sub> + NO" pathway to generate N<sub>2</sub>O and improve the selectivity of N<sub>2</sub>. SO<sub>2</sub>-TPD analyses showed that Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst has better SO<sub>2</sub> resistance with less formation of sulfate species. The NH<sub>3</sub>-TPD, H<sub>2</sub>-TPR and in-situ DRIFTS results indicated that the appropriate amount of Ni modified Ni<sub>1</sub>Mn<sub>0.5</sub>Al<sub>0.5</sub>O<sub>x</sub> catalyst could modulate the surface acidity, redox properties as well as the adsorption and activation active sites. In DFT calculations, charge density difference analysis revealed the

presence of strong electron leaps between atoms as well as intermetallic synergies. The adsorption energy confirmed that Ni greatly enhanced the interaction between NO and Mn sites, prompting the NH<sub>3</sub>-SCR to follow the L-H mechanism more efficiently. In addition, Ni attenuated the strong surface Lewis acid of NiMnAlO<sub>x</sub> catalysts to medium surface Lewis acid and reduced the N<sub>2</sub>O generated by the NH<sub>3</sub> over-oxidation pathway. The d-band PDOS analysis of Mn showed that the Jahn-Teller degradation promoted a more localized d-band and a greater degree of spin-polarization of Mn in NiMnAlO<sub>x</sub>. The adsorption of paramagnetic NO on the NiMnAlO<sub>x</sub> catalyst was accompanied by a magnetic moment reversal in the active center and the catalyst as a whole, which weakened the large magnetic moment of the Mn site and led to the stabilization of the overall antiferromagnetic state of the catalyst. In contrast, the interaction of non-magnetic NH<sub>3</sub> with Mn sites did not undergo magnetic moment reversal and weakening of the magnetic moment of Mn sites, preventing the over-bonding of NH<sub>3</sub> with Mn. In addition, the PDOS of NO/NH<sub>3</sub> adsorption models of the catalysts provided insight into the mechanism and influence of the degradation of the Jahn-Teller effect and the reversal of magnetic moment reversal of Mn sites on the surface adsorption process.

#### CRediT authorship contribution statement

**Qinghua Yan:** Investigation, Writing – original draft. **Jiwen Xiao:** Visualization, Data curation, Validation, DFT calculations. **Rongrong Gui:** Visualization, Formal analysis. **Zhenyu Chen:** Supervision, Project administration. **Yuxin Wang:** Investigation, Software. **Yuran Li:** Validation, Supervision. **Tingyu Zhu:** Resources, Supervision, Project administration. **Qiang Wang:** Conceptualization, Writing – review & editing. **YanJun Xin:** Resources, Writing – review & editing.

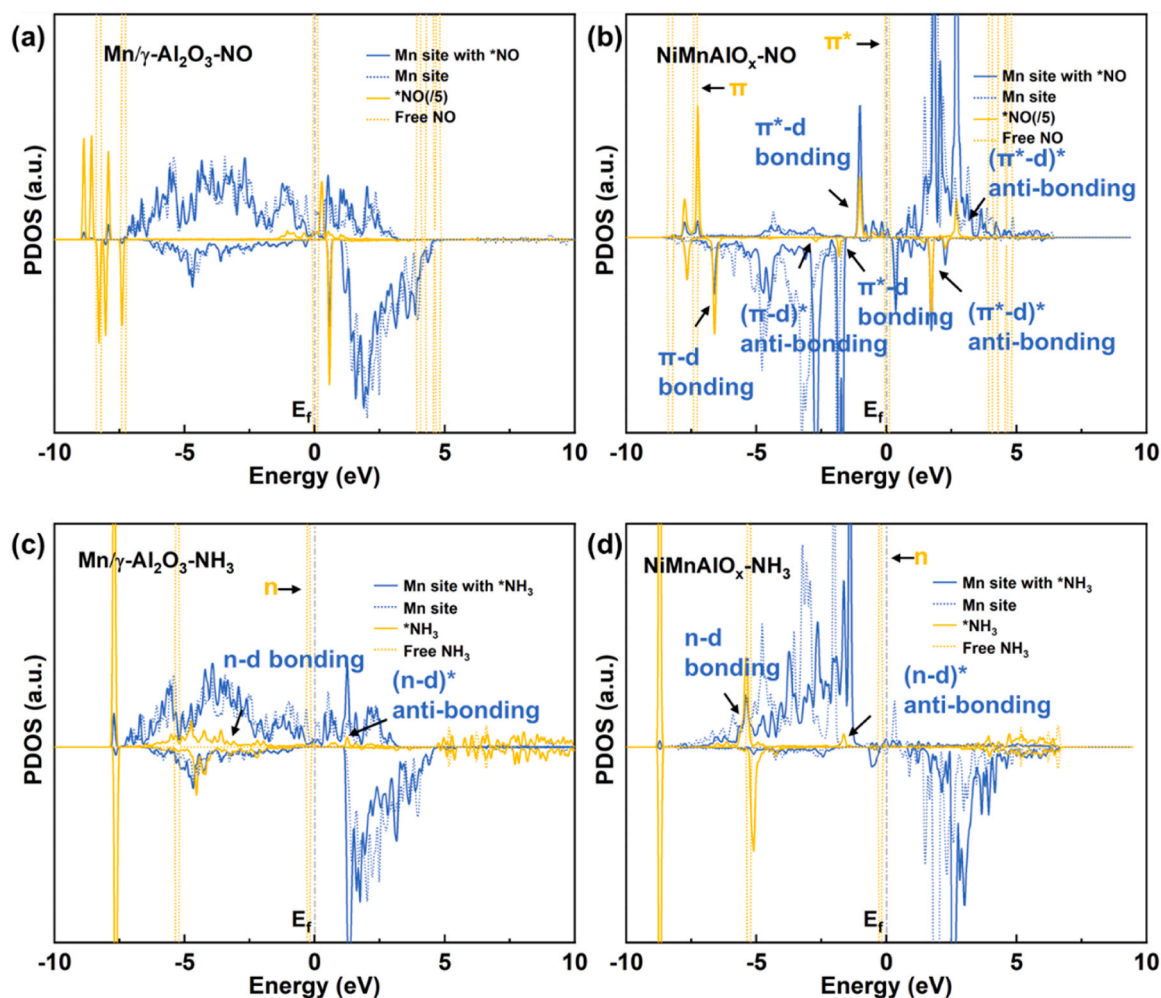


Fig. 11. PDOS profiles of NO and NH<sub>3</sub> adsorbed on (a, c) Mn/γ-Al<sub>2</sub>O<sub>3</sub> and (b, d) NiMnAlO<sub>x</sub>. (Dash lines represents clean surfaces or small molecular and solid lines represents adsorbed small molecular or surfaces, grey dash dot lines mark Fermi level).

### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

### Data availability

Data will be made available on request.

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### Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.apcatb.2023.123489](https://doi.org/10.1016/j.apcatb.2023.123489).

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